

“ABC”-models for subgrain structure evolution in MatCalc 6

(MatCalc 6.00.0258)

P. Warczok



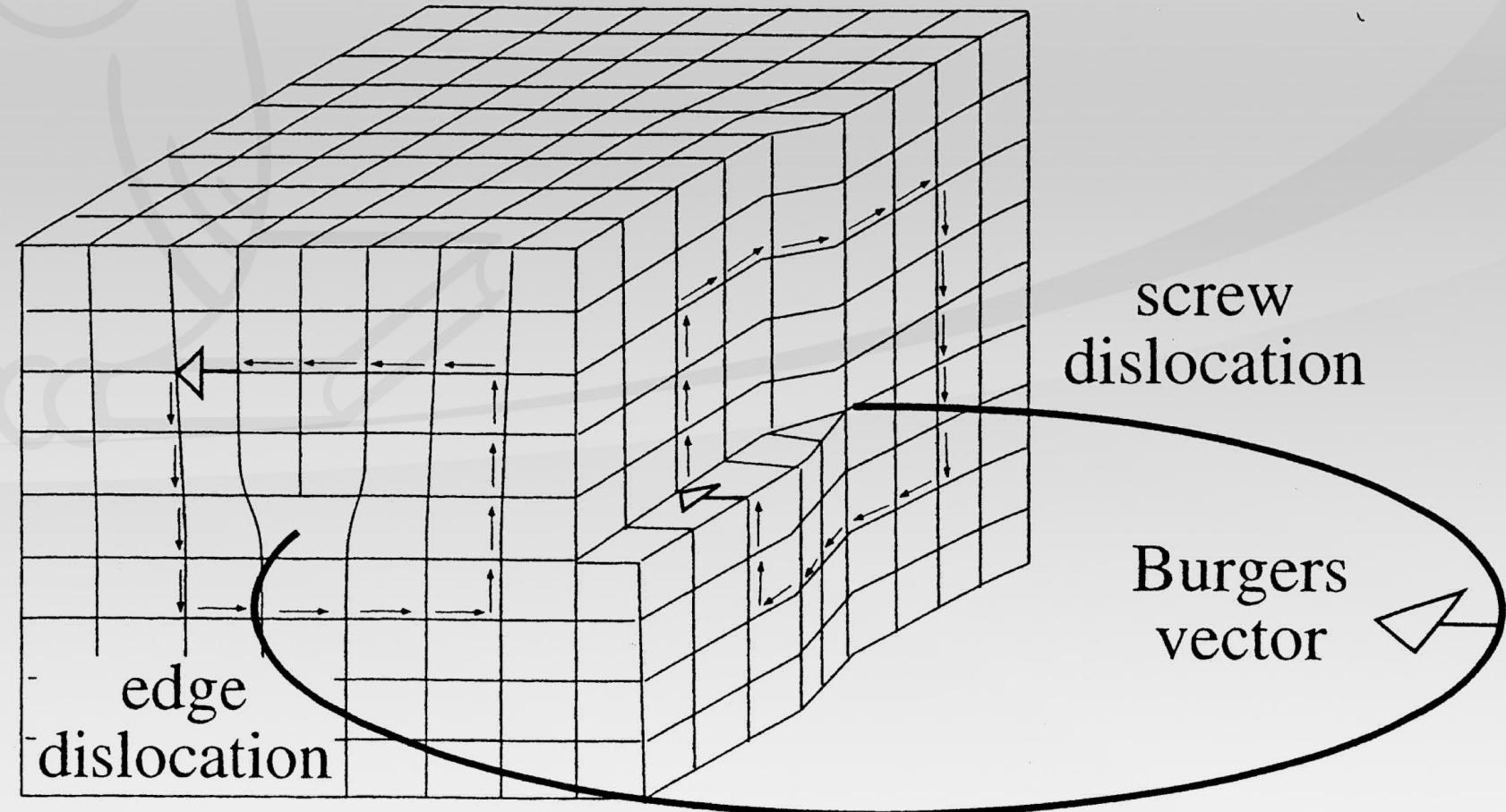
Outlook

- Few words on substructure
- Dislocation density evolution
- Subgrain size evolution
- Model demonstration

Introduction to dislocations

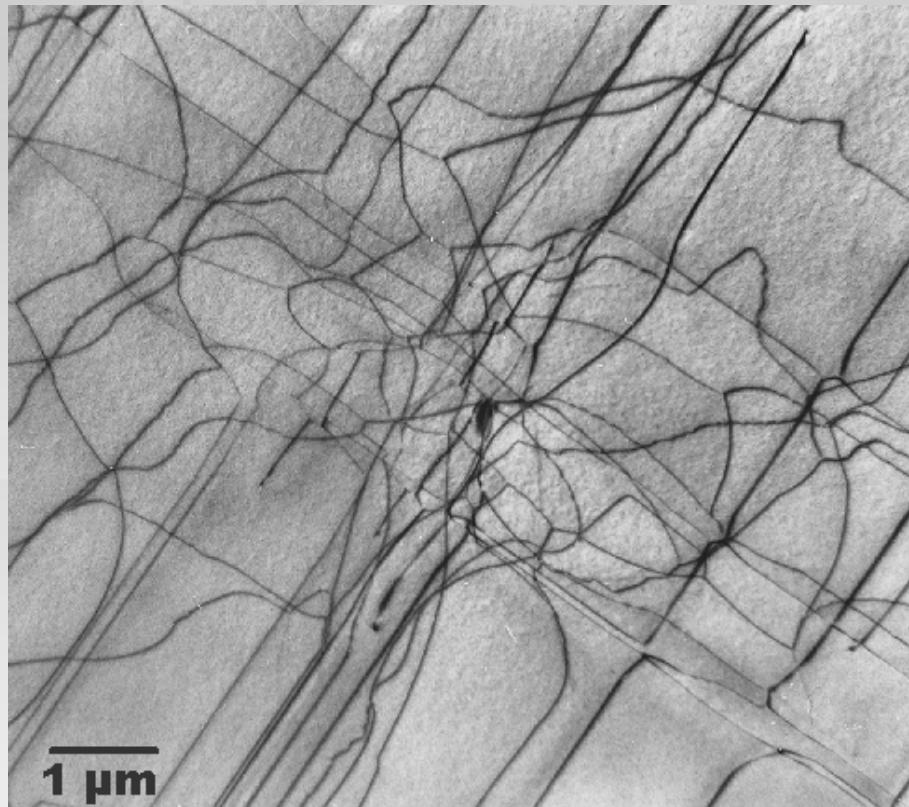
Dislocations

- Two geometries:
 - Edge
 - Screw

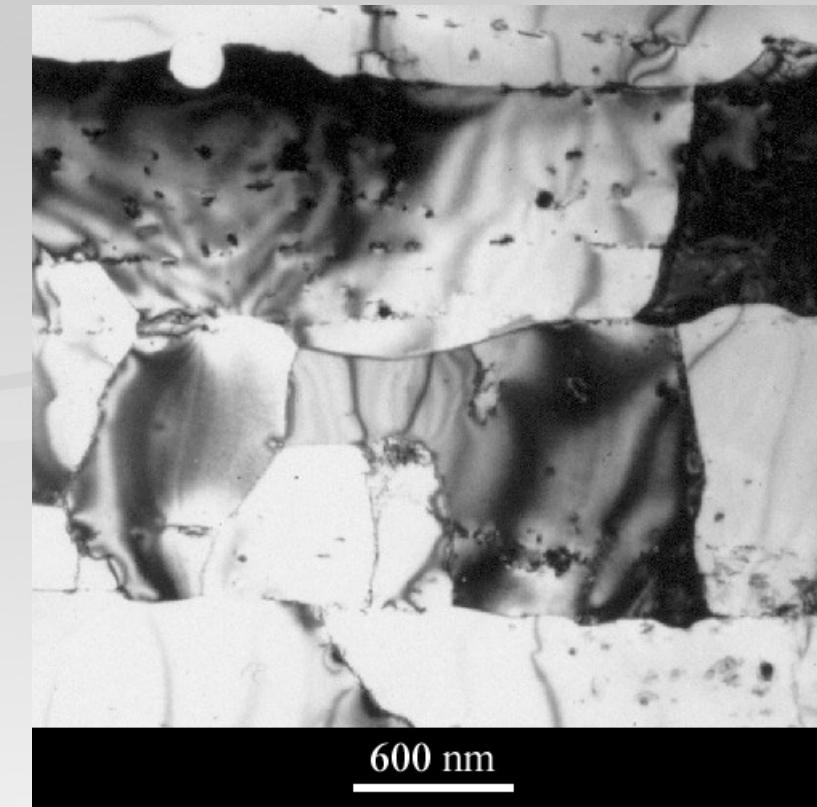


Dislocations

Intrinsic dislocations



Wall dislocations



Dislocation density

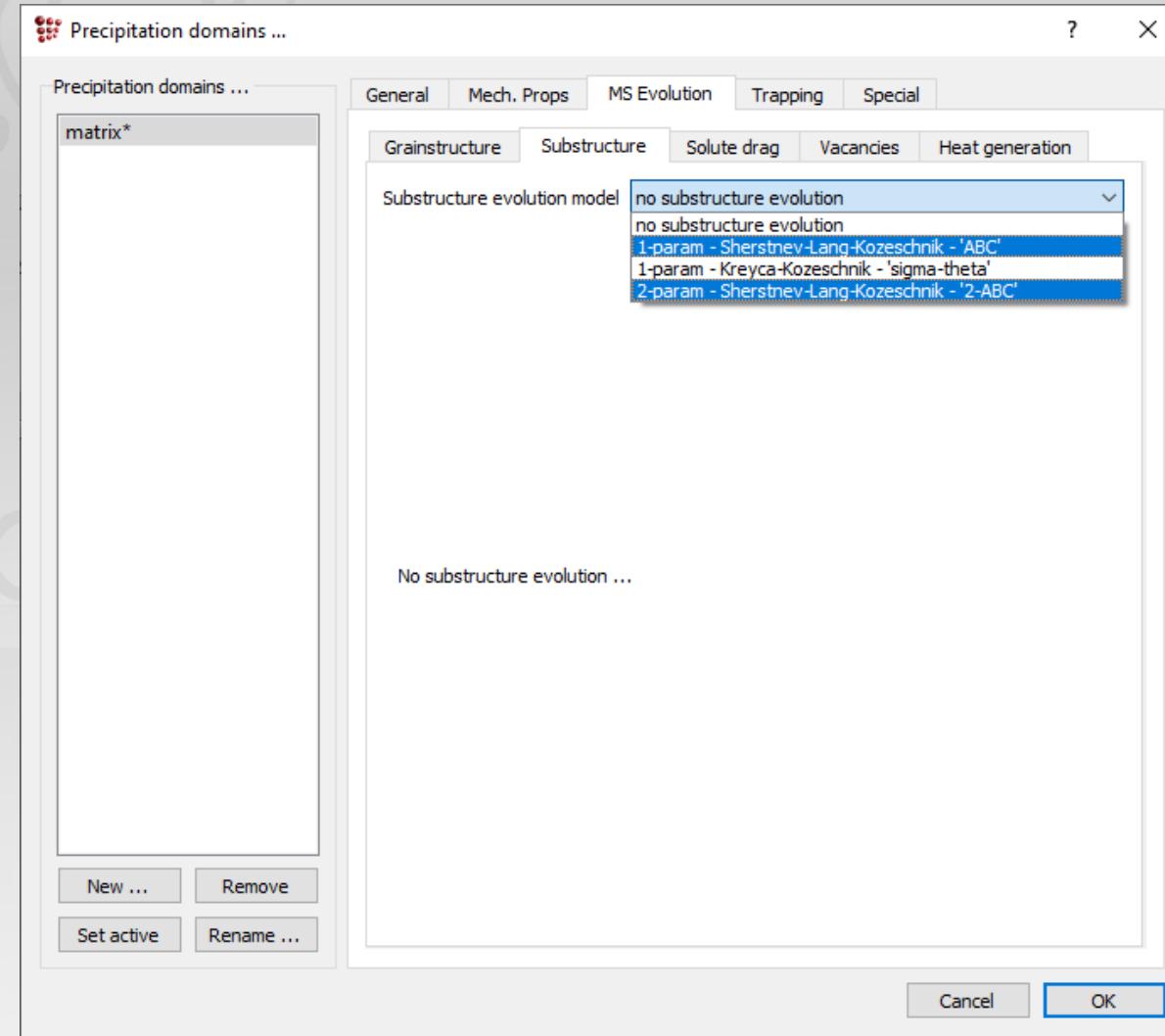
- Impact:
 - Diffusion (pipe-diffusion)
 - Nucleation rate (number of nucleation sites)
 - Subgrain size (through similitude principle)
 - Recrystallization onset
 - Yield strength
 - Directly - work hardening
 - Indirectly – subgrain size, precipitate size

Dislocation density evolution

Dislocation density evolution

- MatCalc models
 - Sherstnev-Lang-Kozeschnik (SLK) models
 - 1 parameter model (a.k.a. „1ABC“)
 - 2 parameters model (a.k.a. „2ABC“)
 - 1 parameter model: global dislocation density evolution
 - 2 parameters model: separate dynamics for intrinsic and wall dislocations

Dislocation density evolution



Dislocation density evolution

$$\dot{\rho} = \dot{\rho}_1 - \dot{\rho}_2 - \dot{\rho}_3$$

- Dislocation generation
 - Deformation $\rightarrow \dot{\rho}_1$
- Dislocation annihilation
 - Dynamic recovery (dislocations with antiparallel Burgers vectors hit each other) $\rightarrow \dot{\rho}_2$
 - Static recovery (dislocation climb) $\rightarrow \dot{\rho}_3$

Dislocation generation (deformation)

$$\dot{\rho}_1 = \frac{M \dot{\varepsilon}}{bA} \sqrt{\rho}$$

ρ - Dislocation density

A - A-parameter (constant)

M - Taylor factor

$\dot{\varepsilon}$ - Strain rate

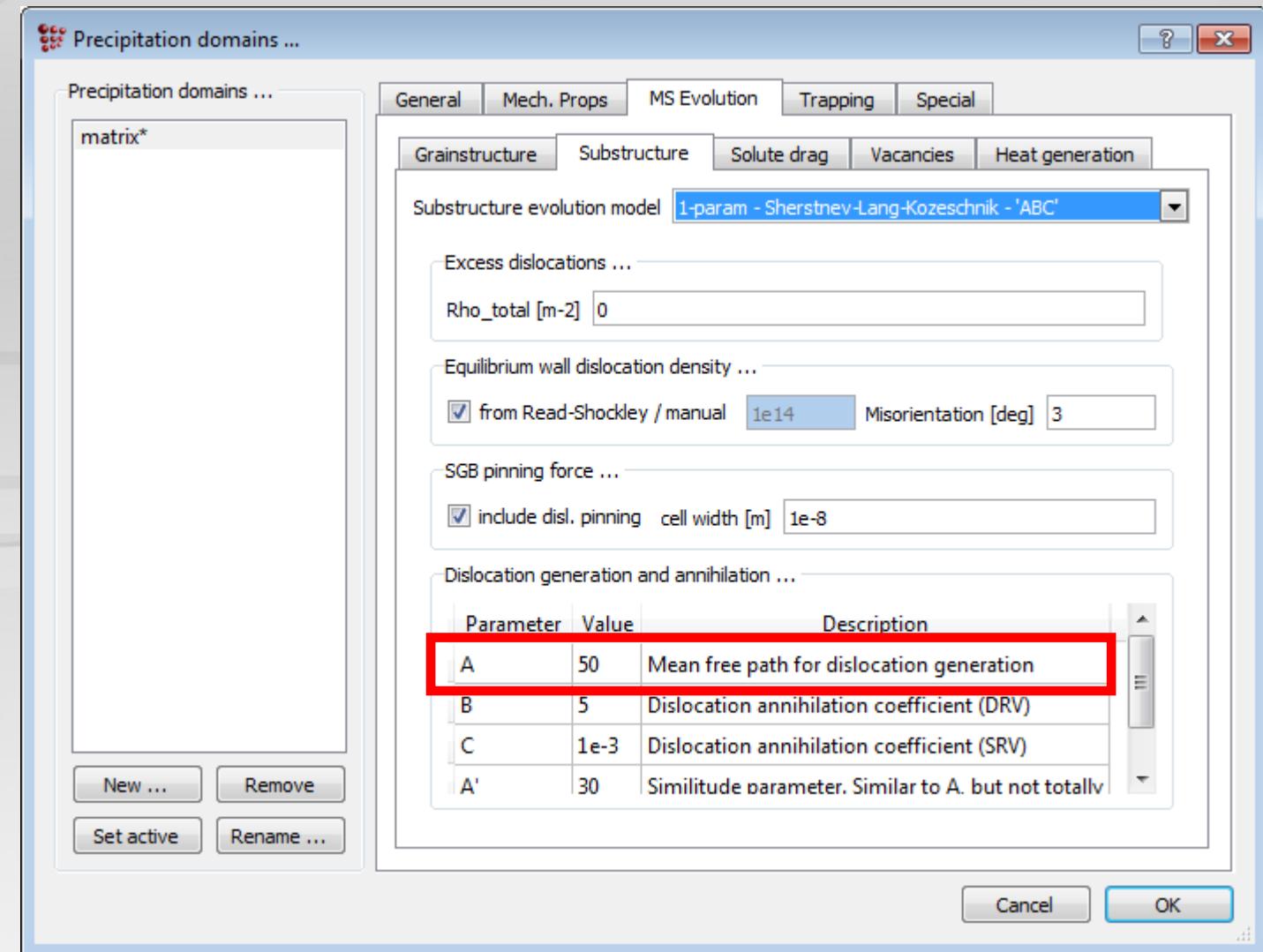
b - Burgers vector

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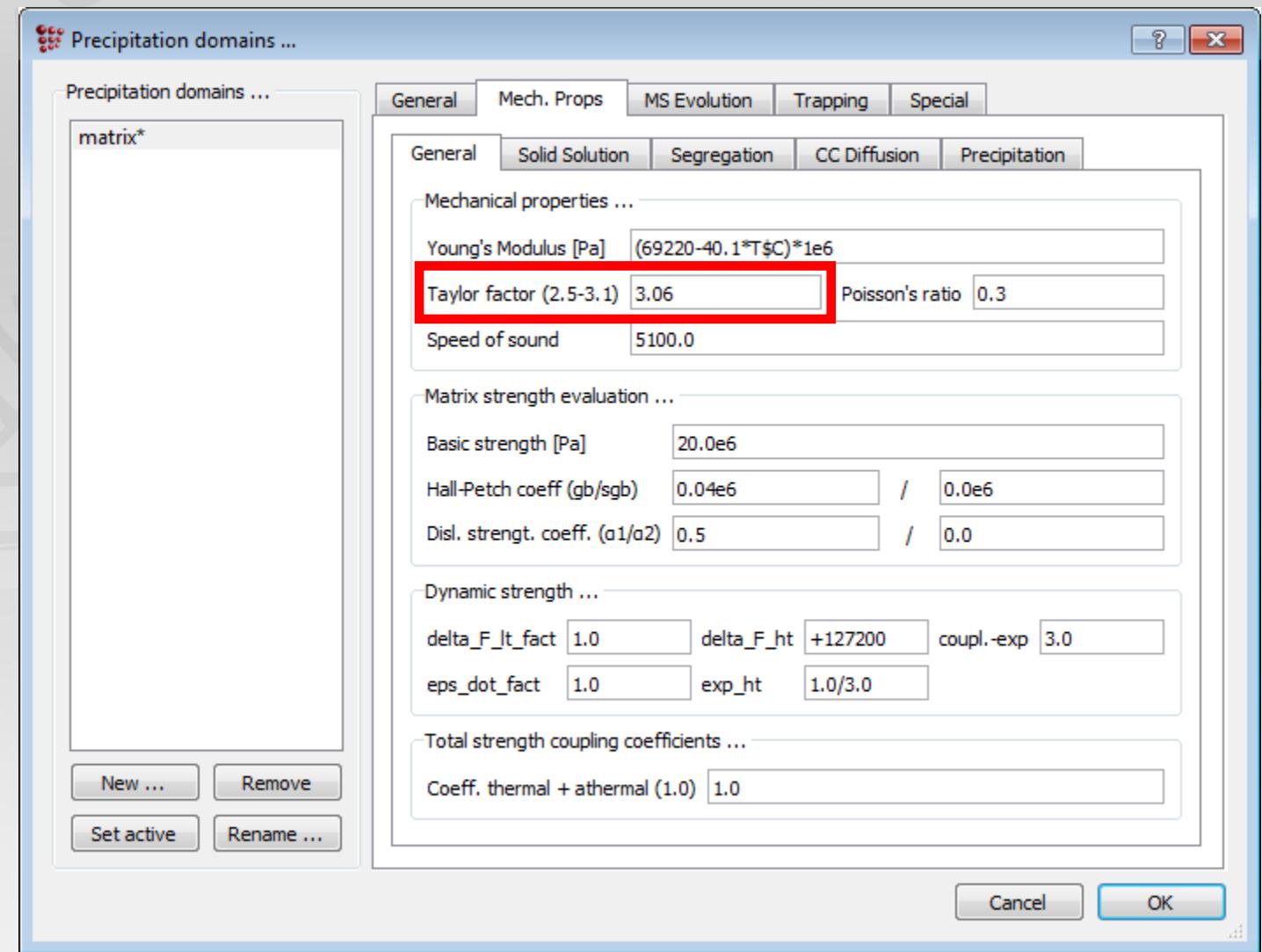
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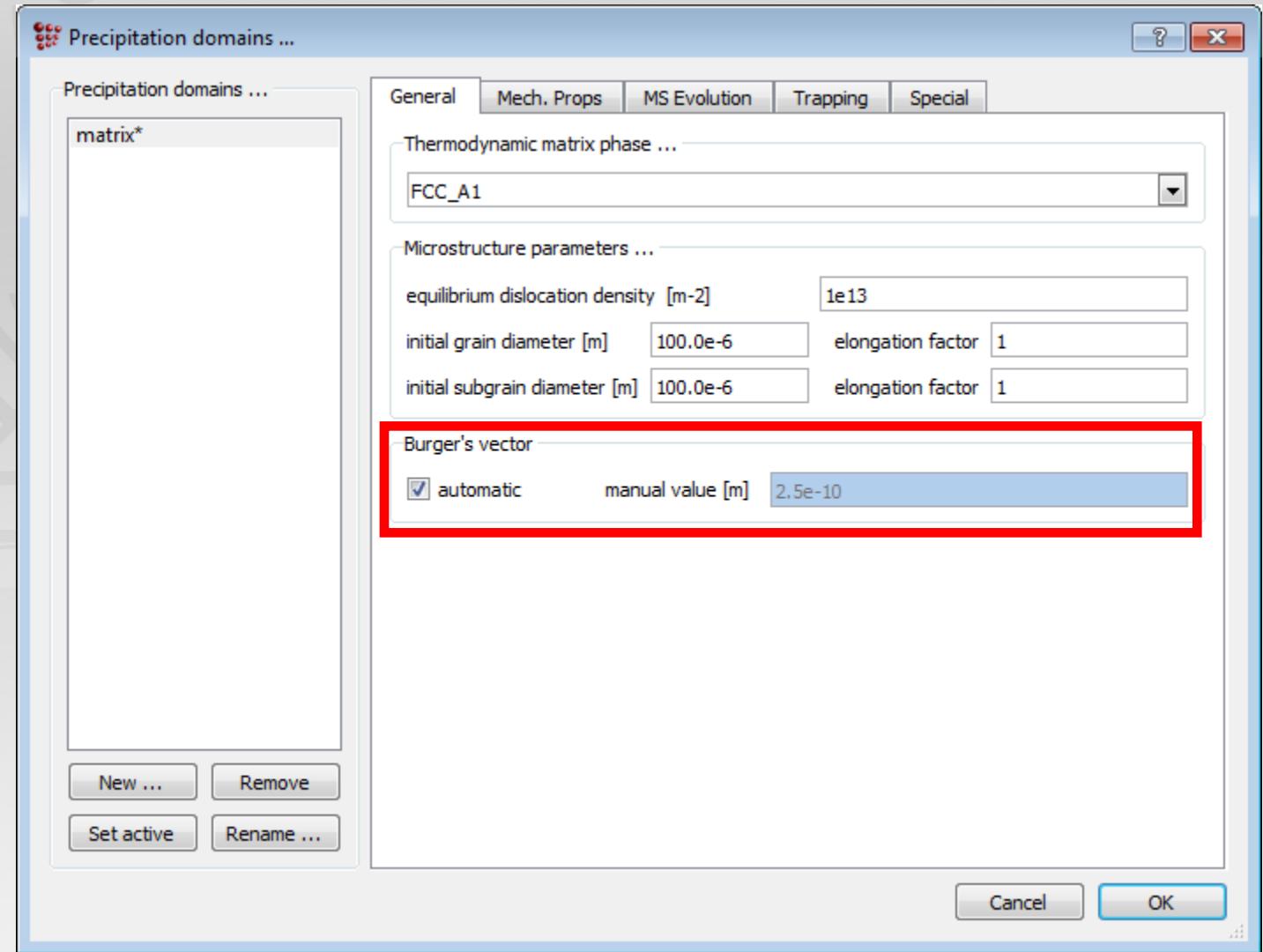
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Dislocation generation (deformation)

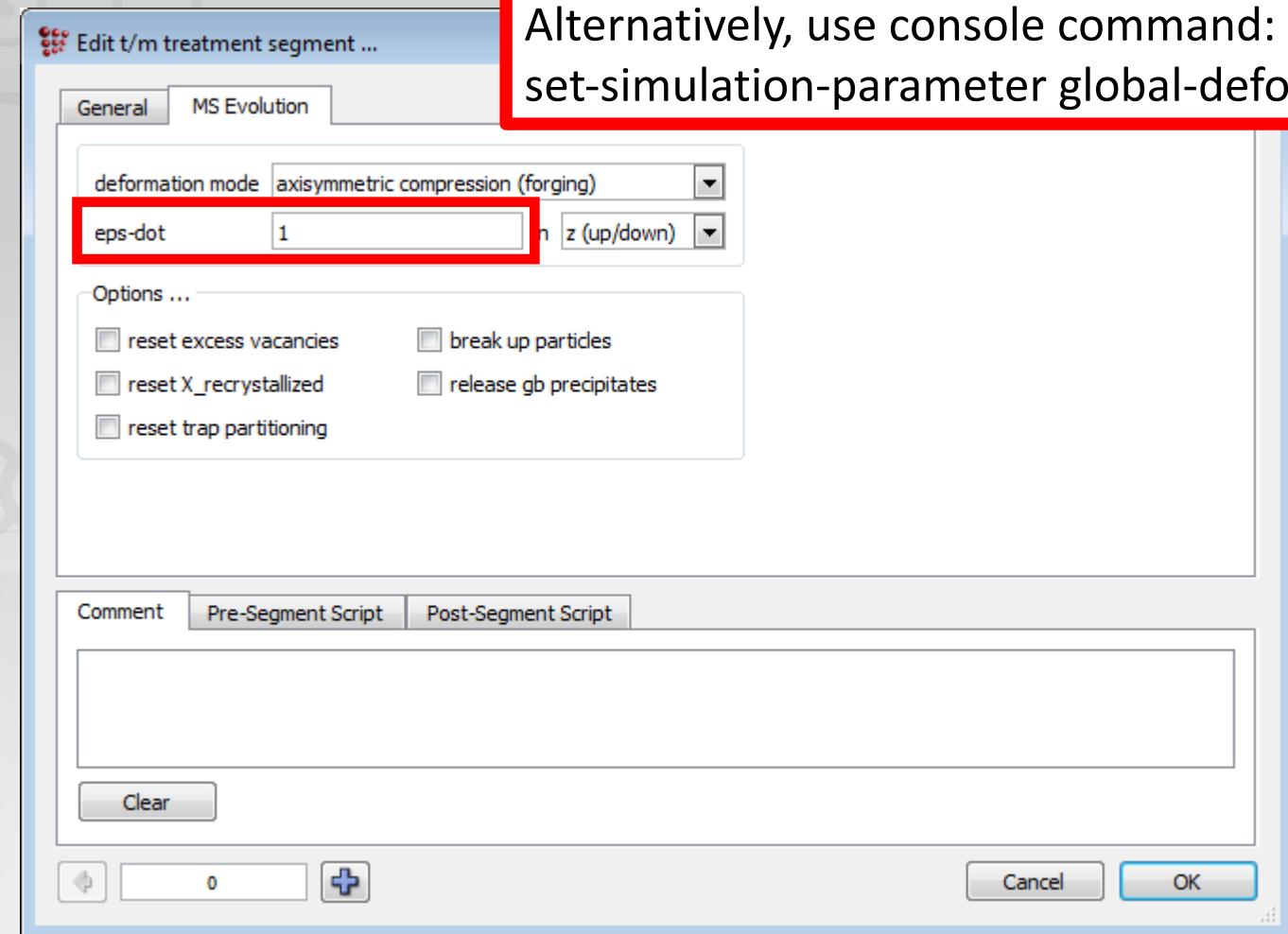
$$\dot{\rho}_1 = \frac{M \dot{\varepsilon}}{bA} \sqrt{\rho}$$

M - Taylor factor

$\dot{\varepsilon}$ - Strain rate

b - Burgers vector

Alternatively, use console command:
set-simulation-parameter global-deformation-rate=1



Dislocation annihilation (dynamic recovery)

$$d_{ann} = \frac{Gb^4 N_A}{2\pi(1 - \nu)E_{Va}}$$

d_{ann} - Annihilation distance

E_{Va} - Vacancy formation energy

G - Shear modulus

(from thermodynamic database)

ν - Poisson ratio

N_A - Avogadro constant

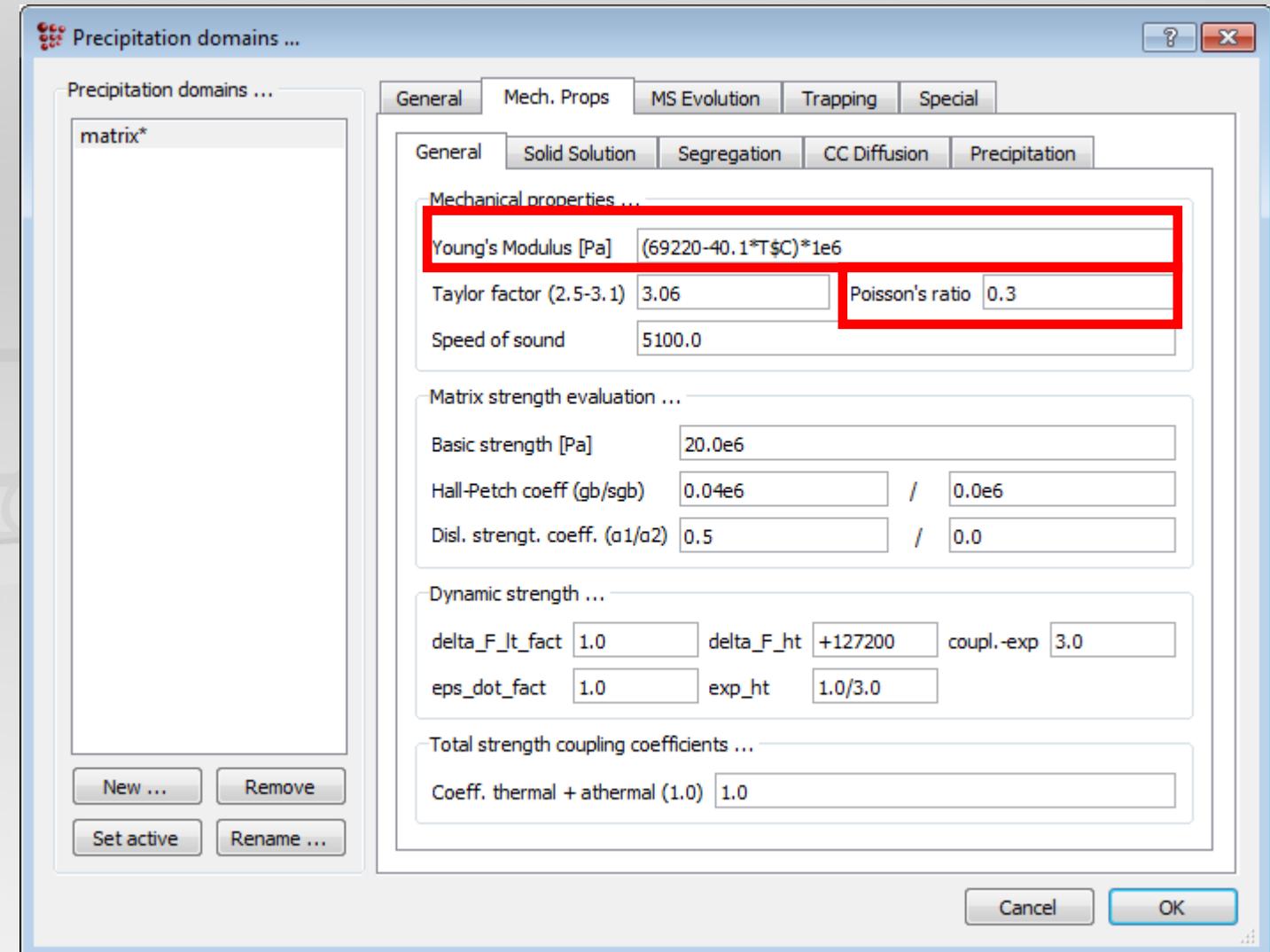
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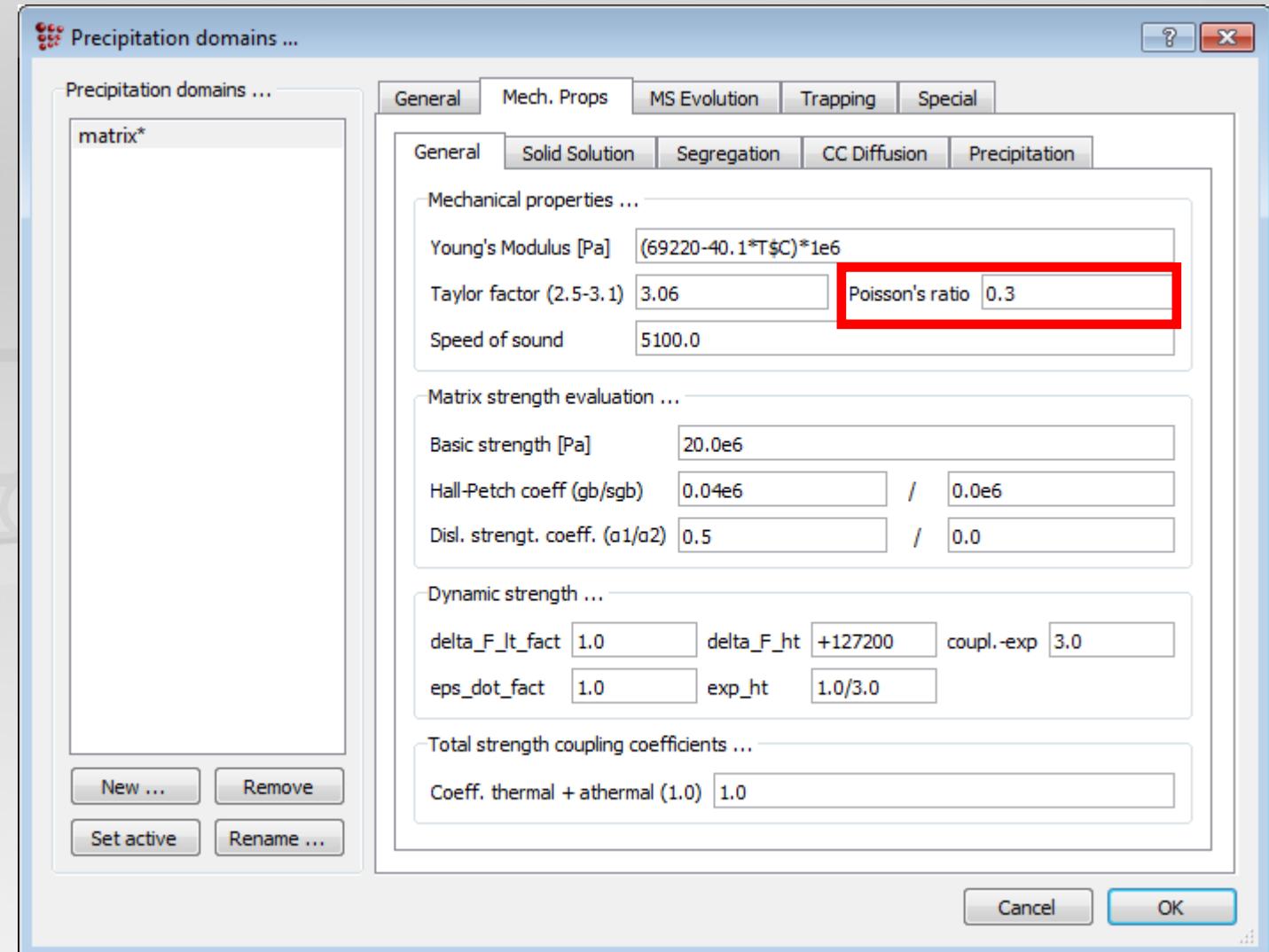
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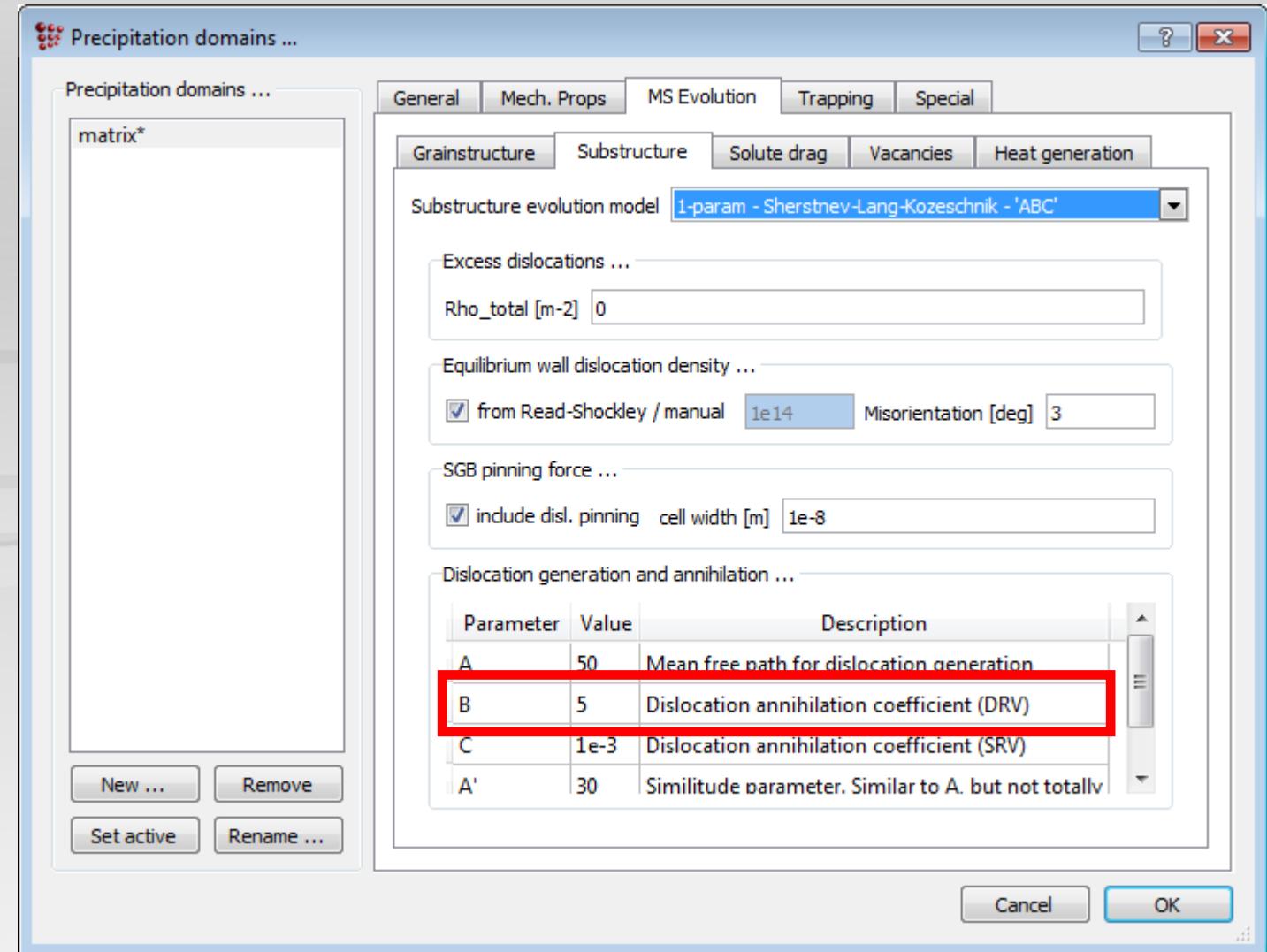
ν - Poisson ratio



Dislocation annihilation (dynamic recovery)

$$\dot{\rho}_2 = \frac{2M\dot{\varepsilon}d_{ann}B}{b}\rho$$

B - B-parameter (constant)



Dislocation annihilation (static recovery)

$$\dot{\rho}_3 = \frac{2Gb^3 D_{eff} C}{k_B T} (\rho^2 - \rho_{eq}^2)$$

D_{eff} - Diffusion, incl. enhancement factors like pipe diffusion,
excess vacancies, etc.

k_B - Boltzmann constant

T - Temperature

C - C-parameter (constant)

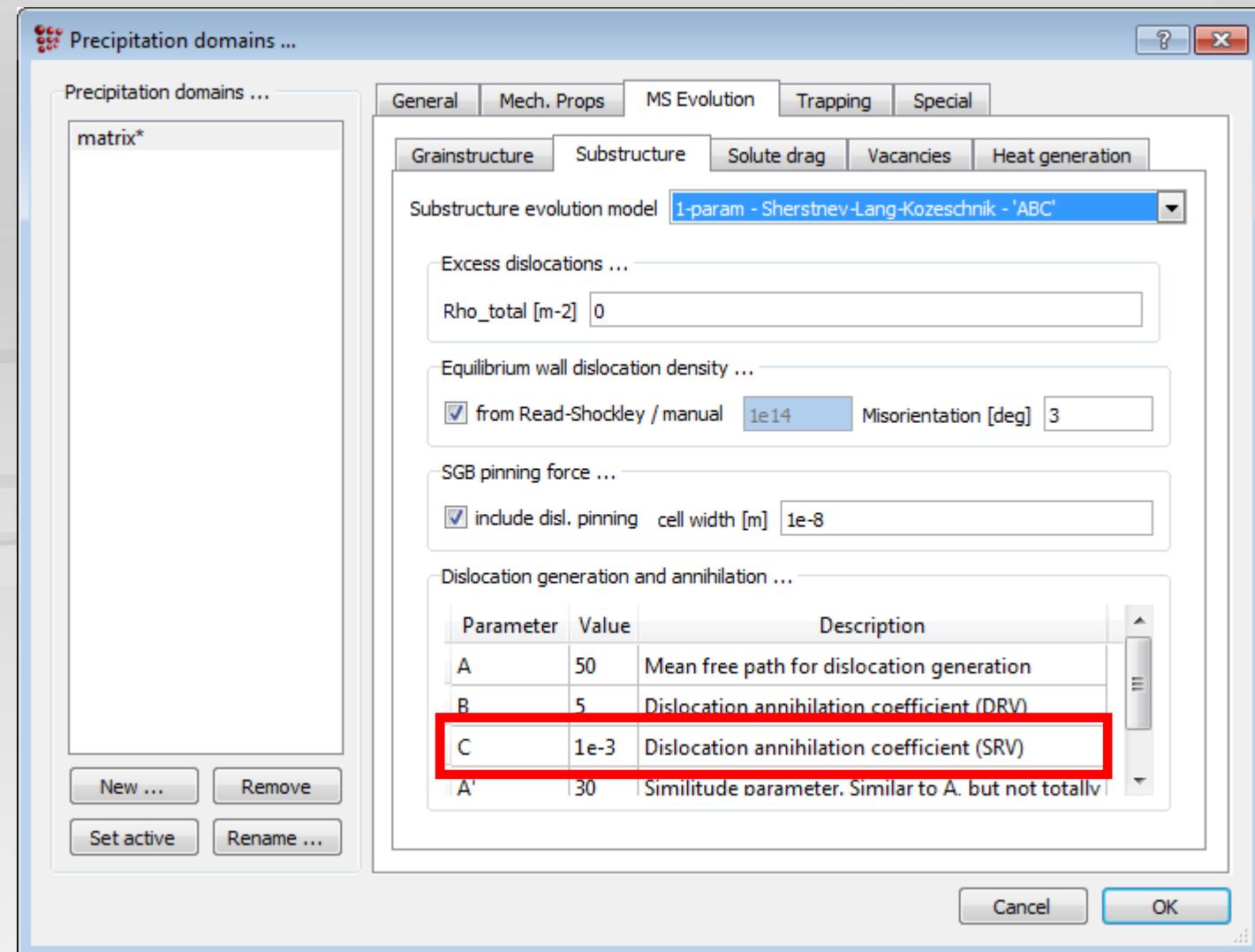
ρ_{eq} - Equilibrium dislocation density

Dislocation annihilation (static recovery)

$$\dot{\rho}_3 = \frac{2Gb^3 D_{eff} C}{k_B T} (\rho^2 - \rho_{eq}^2)$$

C - C-parameter (constant)

ρ_{eq} - Equilibrium dislocation density (sum of intrinsic and wall dislocations)

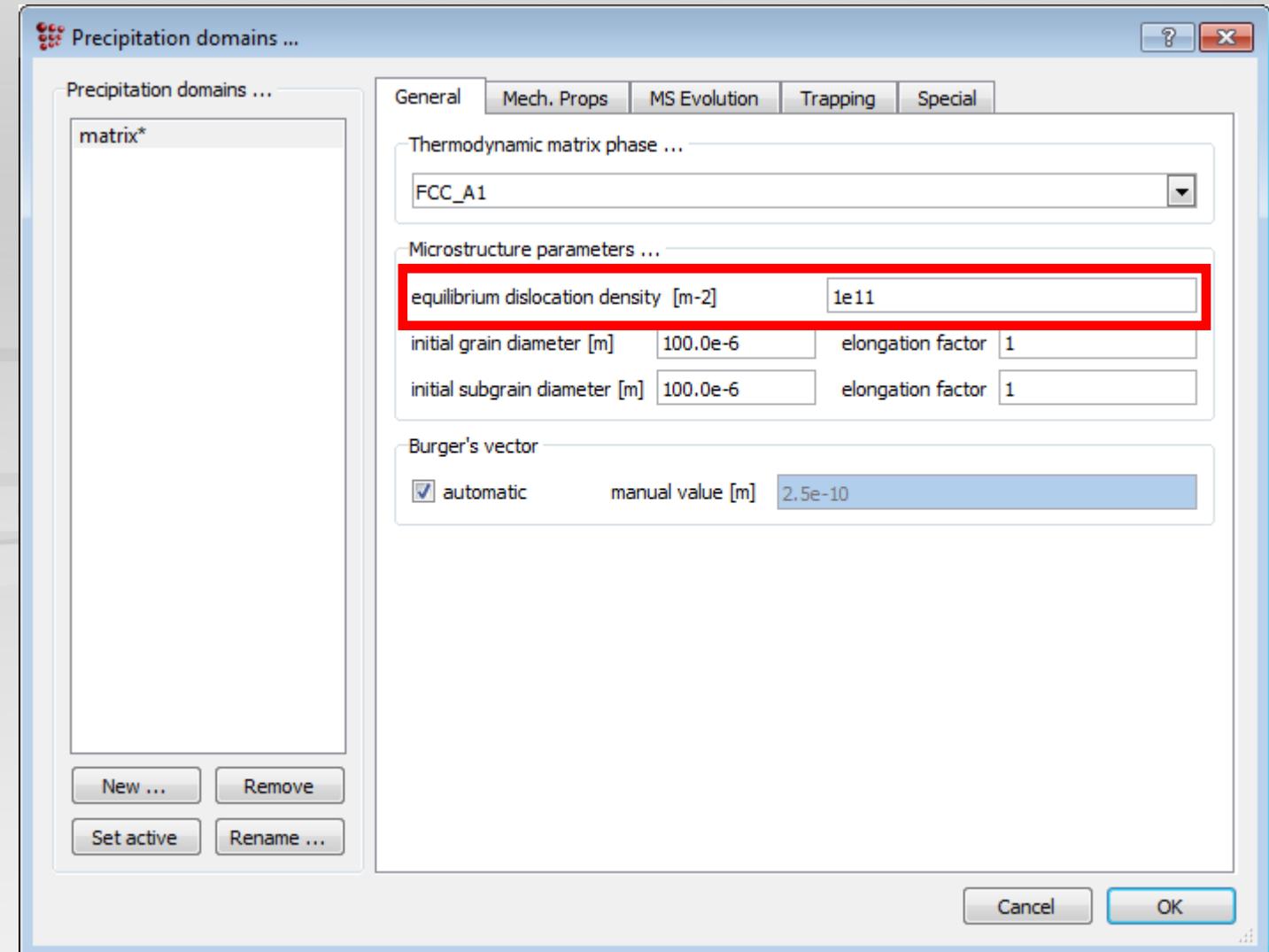


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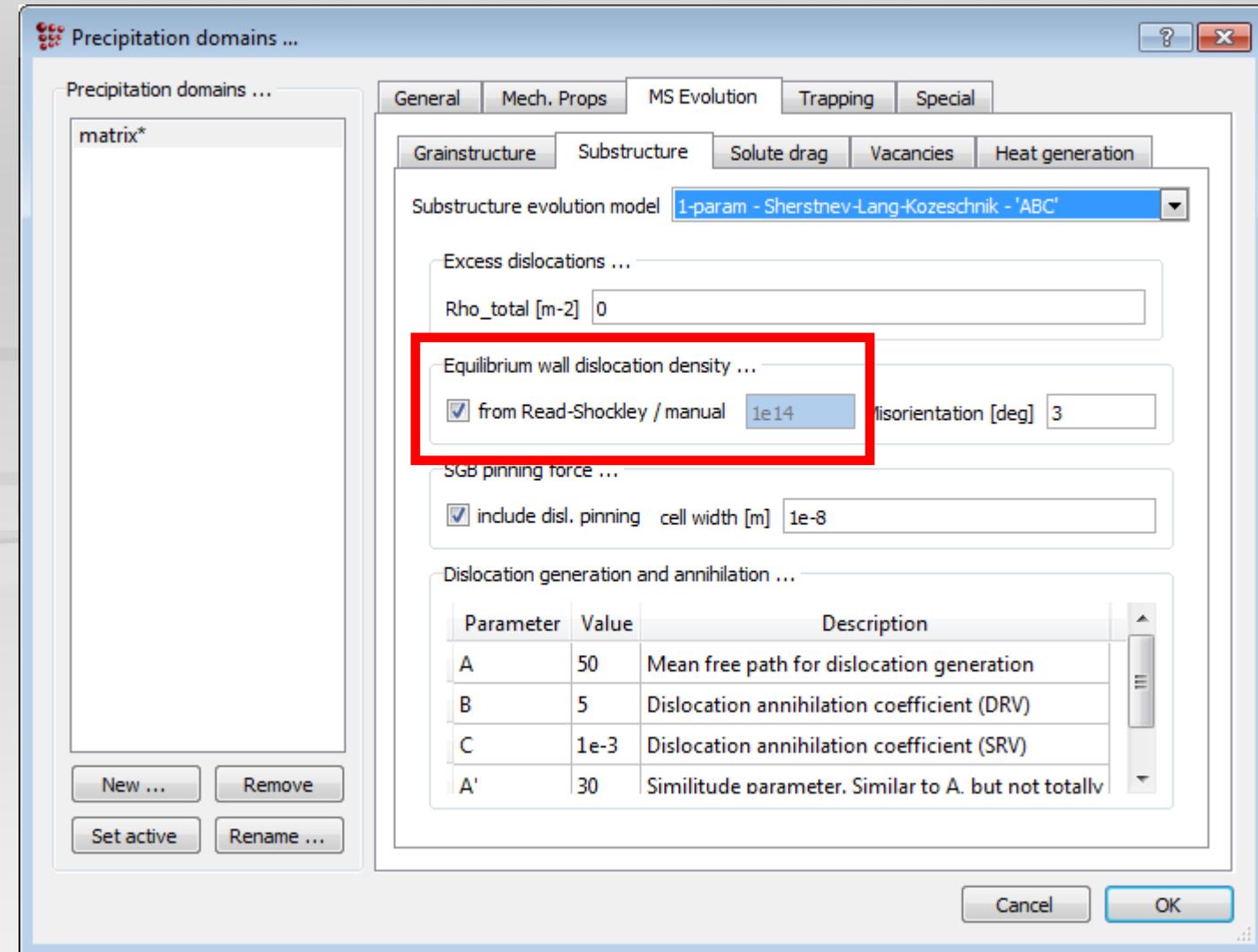


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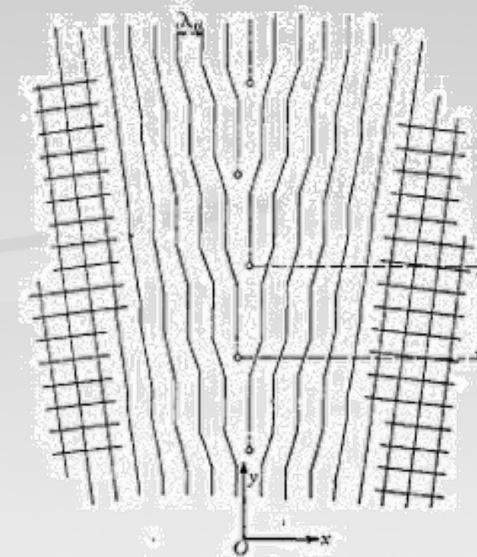
ρ_{eq} - Equilibrium dislocation density (sum of intrinsic and wall dislocations)



Read-Shockley dislocation density

- Read-Shockley dislocation density → necessary amount to fulfill geometrical constraint

$$\rho_{RS} = \frac{\tan\theta}{\delta b}$$



θ - Misorientation angle

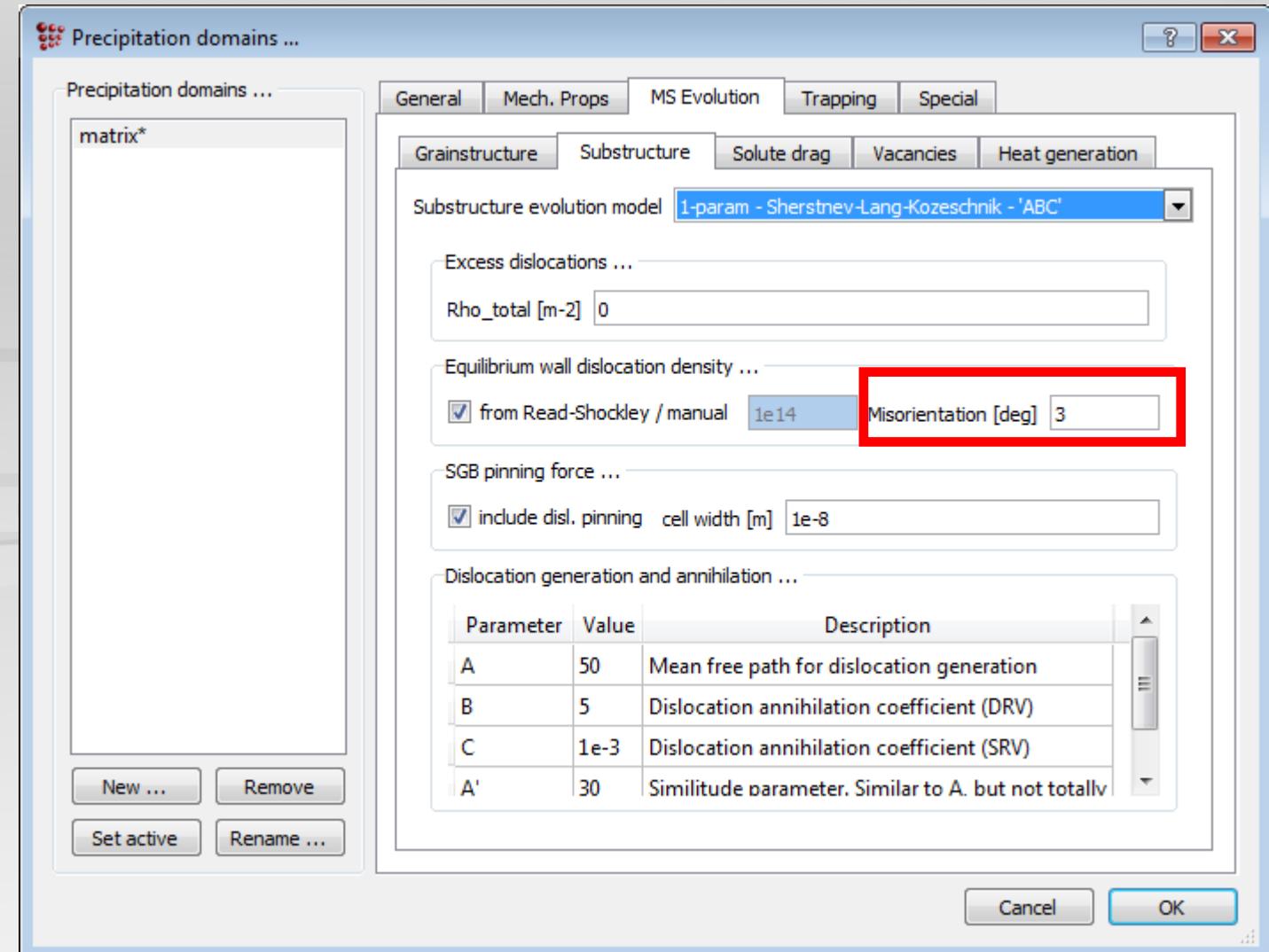
Burgers, J.M., Proc. Phys. Soc. 52 (1940) 23-33

Read-Shockley dislocation density

- Read-Shockley dislocation density → necessary amount to fulfill geometrical constraint

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Subgrain size evolution

Subgrain size evolution

$$\dot{\delta} = \dot{\delta}_1 - \dot{\delta}_2$$

- Subgrains grow to minimize the subgrain boundary area (minimize the boundary energy) $\rightarrow \dot{\delta}_1$
- Subgrain walls shrink with increasing dislocation density (more wall dislocations available) $\rightarrow \dot{\delta}_2$

Subgrain growth

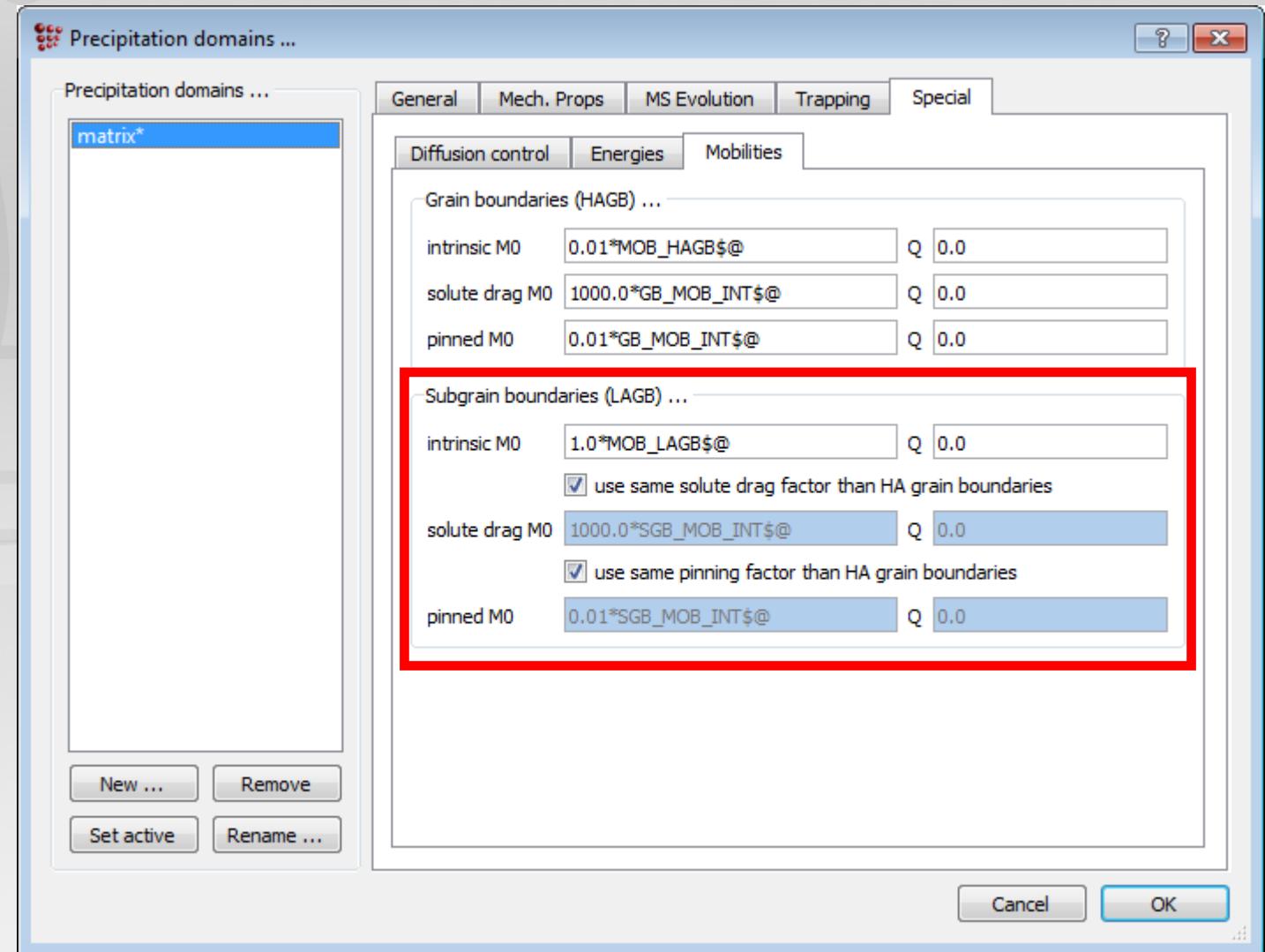
$$\dot{\delta_1} = MP_D$$

- Subgrain growth model same as for grain growth → product of mobility and driving force
- Same models for as for grain boundary mobility → same effects for precipitate pinning and solute drag

Subgrain growth

$$\dot{\delta}_1 = M P_D$$

- Same models as for grain boundary mobility → same effects for precipitate pinning and solute drag



Subgrain growth

- Driving force – balance between Laplace pressure and dislocation pinning of subgrain walls

$$P_D = \frac{4\gamma_{sgb}}{\delta} - \frac{Gb^2}{\sqrt{w\rho}} \sqrt{\rho - \rho_{RS}}$$

γ_{sgb} - Subgrain boundary energy

δ - Subgrain size

w - Cell width for dislocation pinning

ρ_{RS} - Read-Shockley dislocation density

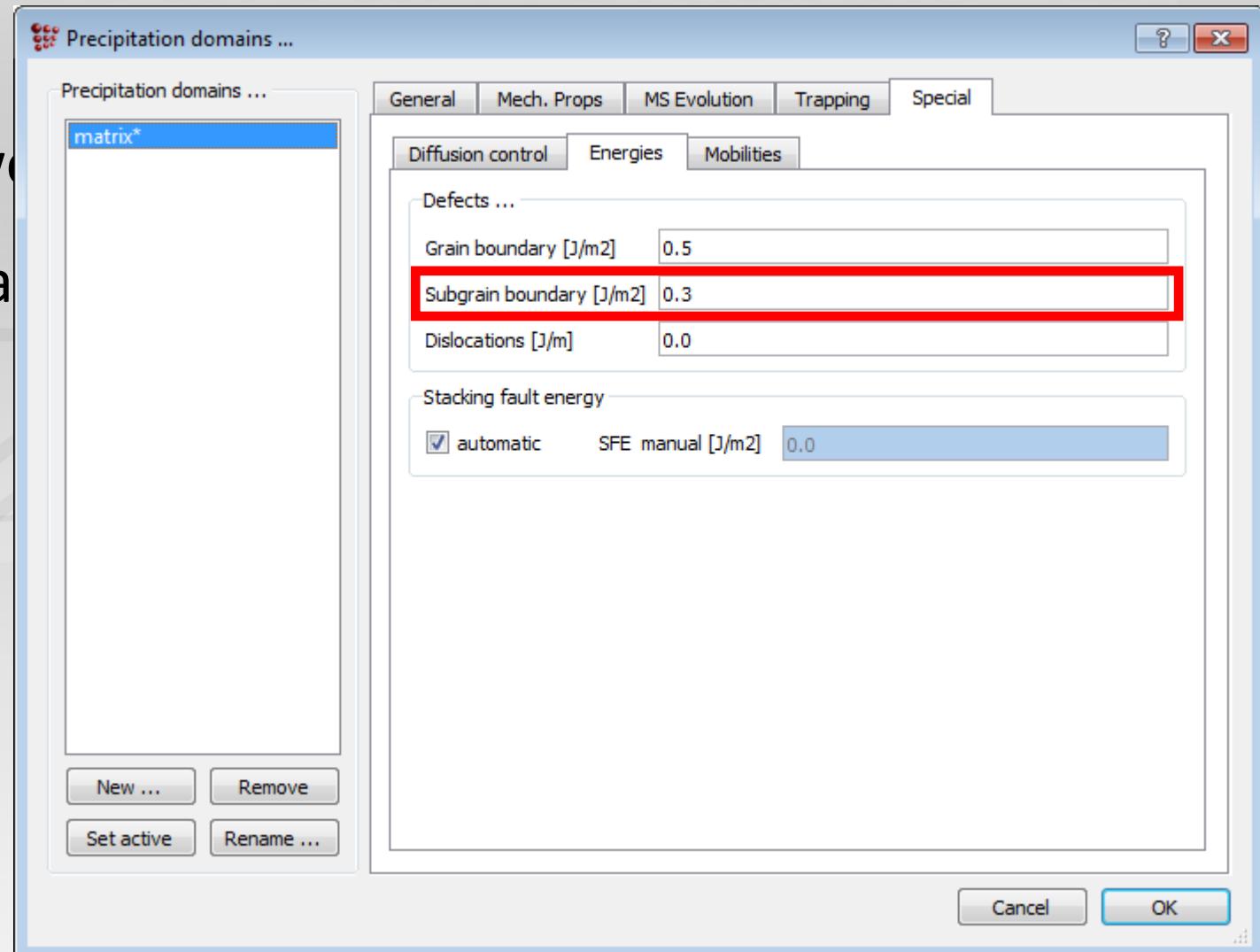
Subgrain growth

- Driving force – balance between dislocation pinning of subgrains

$$P_D = \frac{4\gamma_{sgb}}{\delta} - \frac{Gb^2}{\sqrt{wp}} \sqrt{\rho - \rho_{RS}}$$

γ_{sgb} - Subgrain boundary energy

δ - Subgrain size

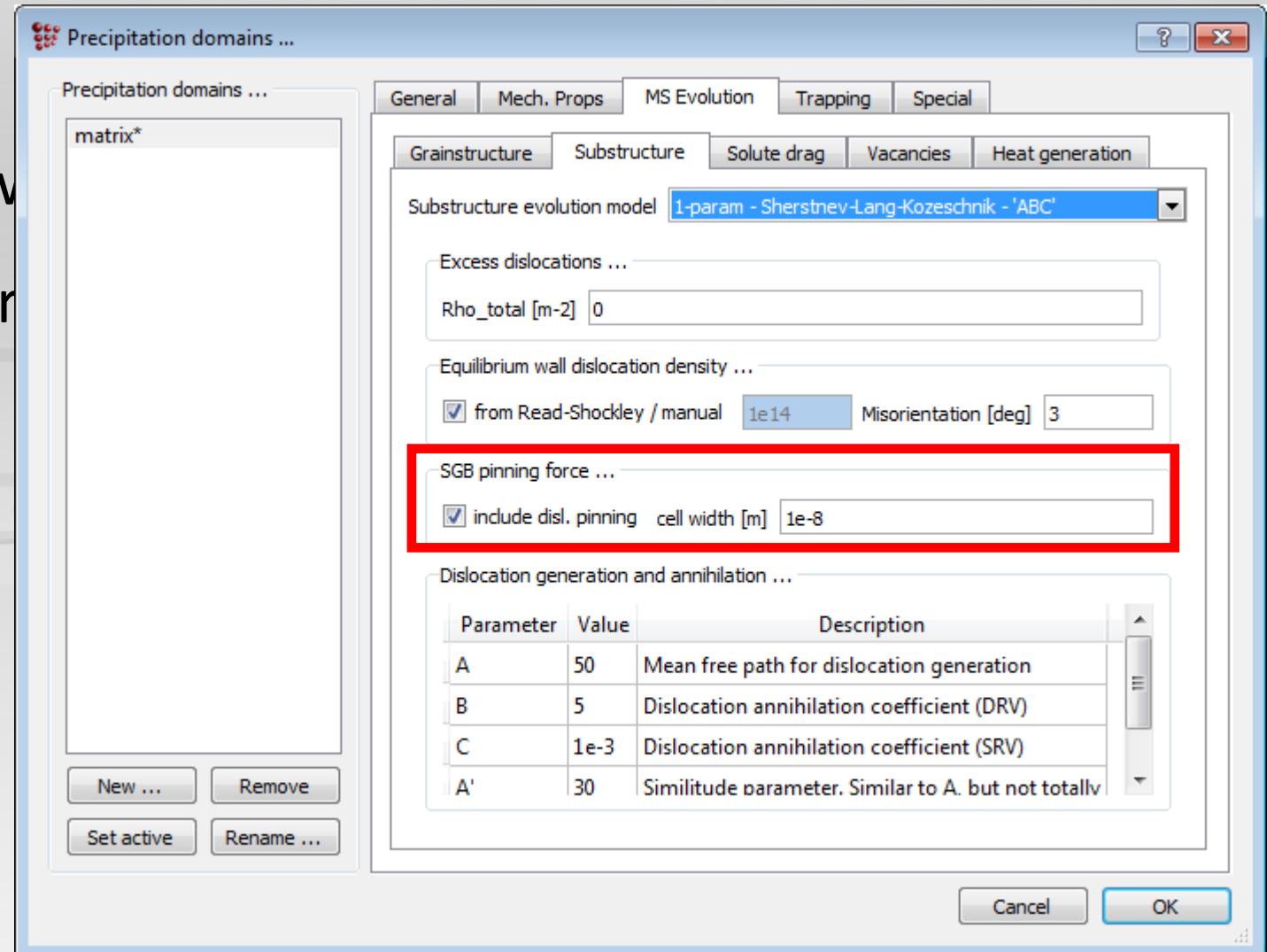


Subgrain growth

- Driving force – balance between dislocation pinning of subgrains

$$P_D = \frac{4\gamma_{sgb}}{\delta} - \frac{Gb^2}{\sqrt{wP}} \sqrt{\rho - \rho_{RS}}$$

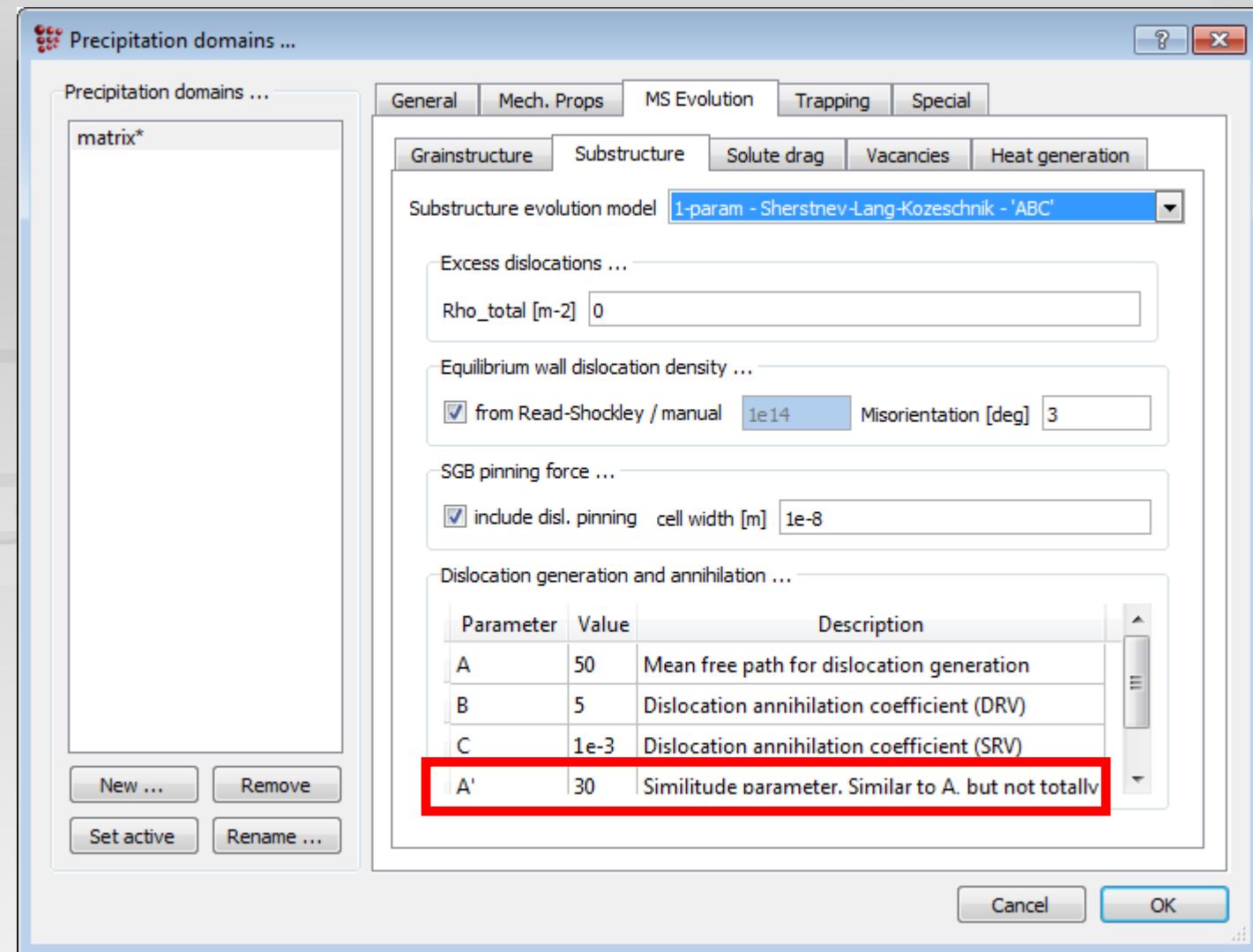
w - Cell width for dislocation pinning



Subgrain shrinkage

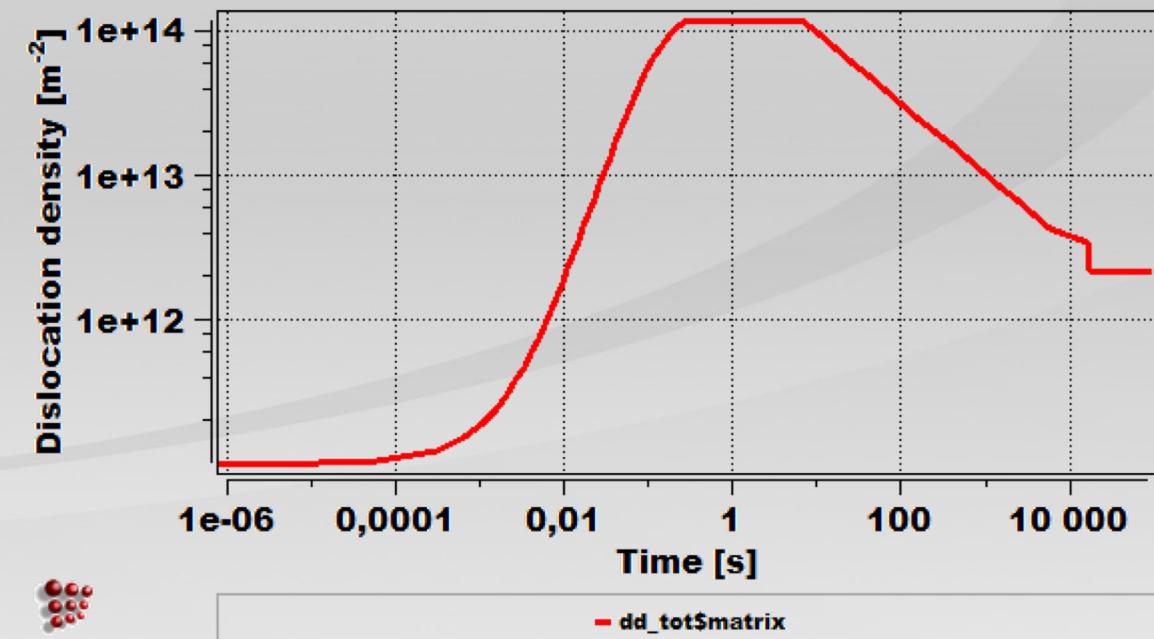
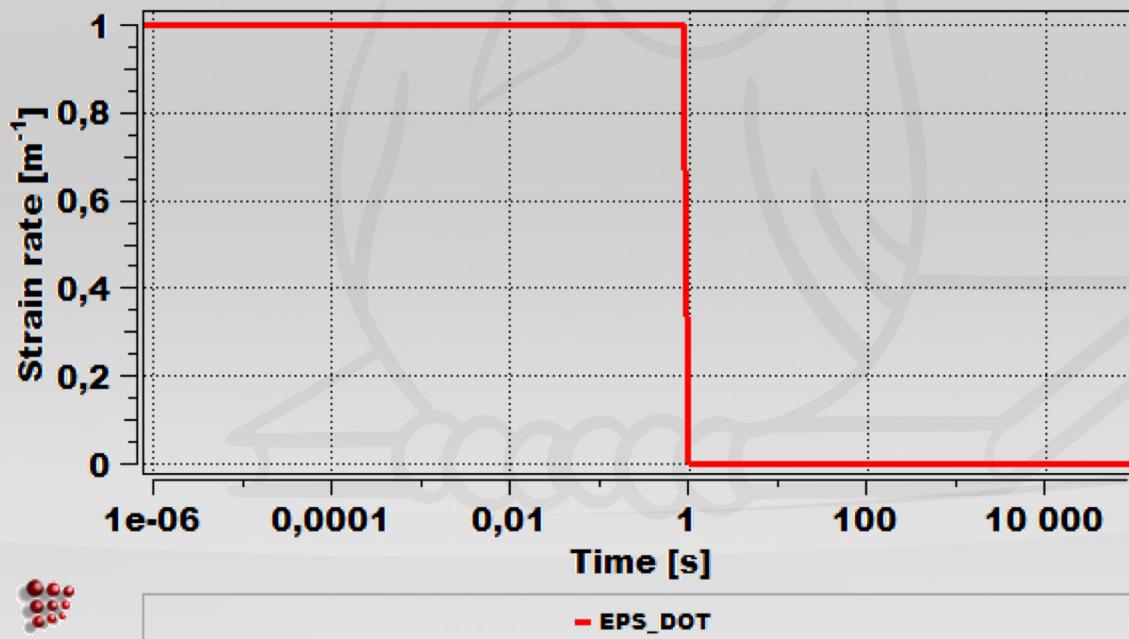
$$\dot{\delta}_2 = \frac{\delta^3}{2(A')^2} \dot{\rho}_1$$

A' - A'-parameter (constant)



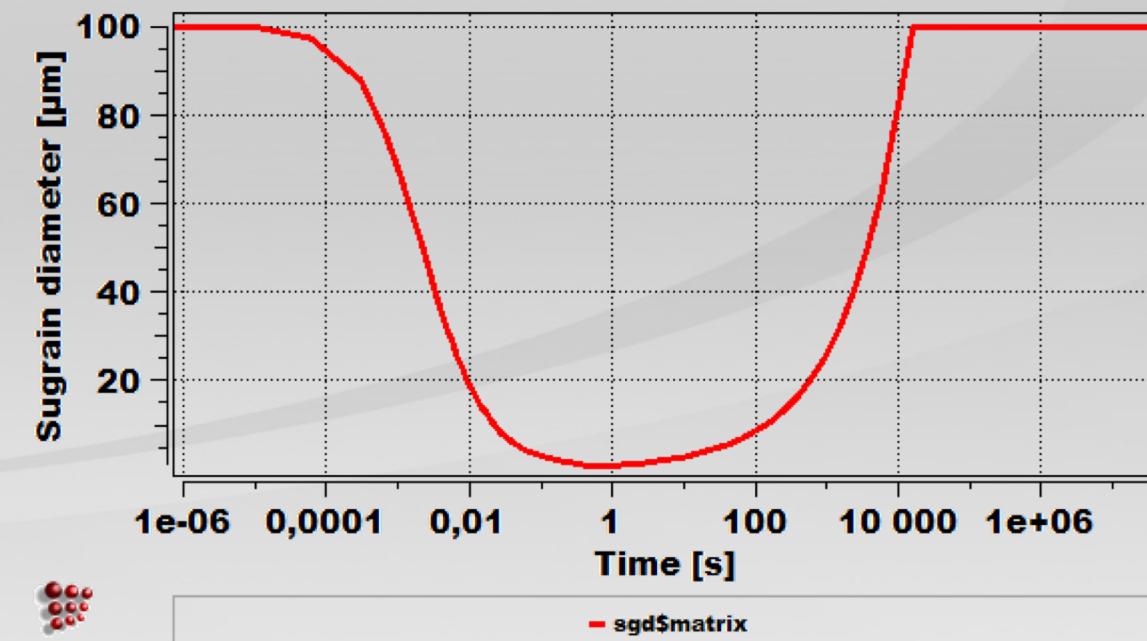
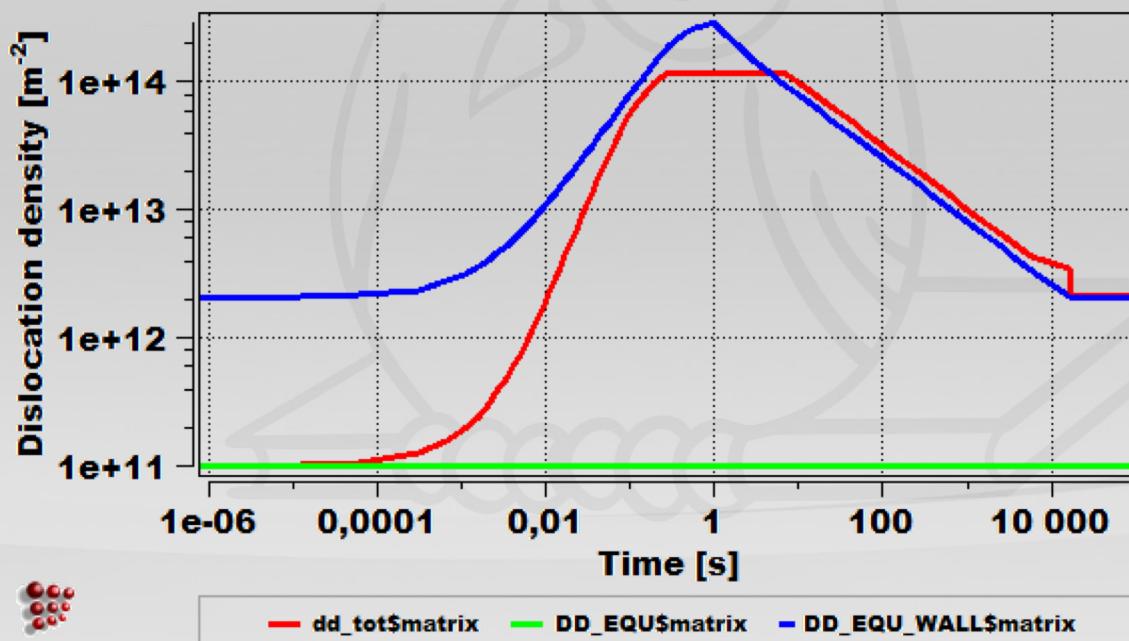
Model demonstration

1-parameter model



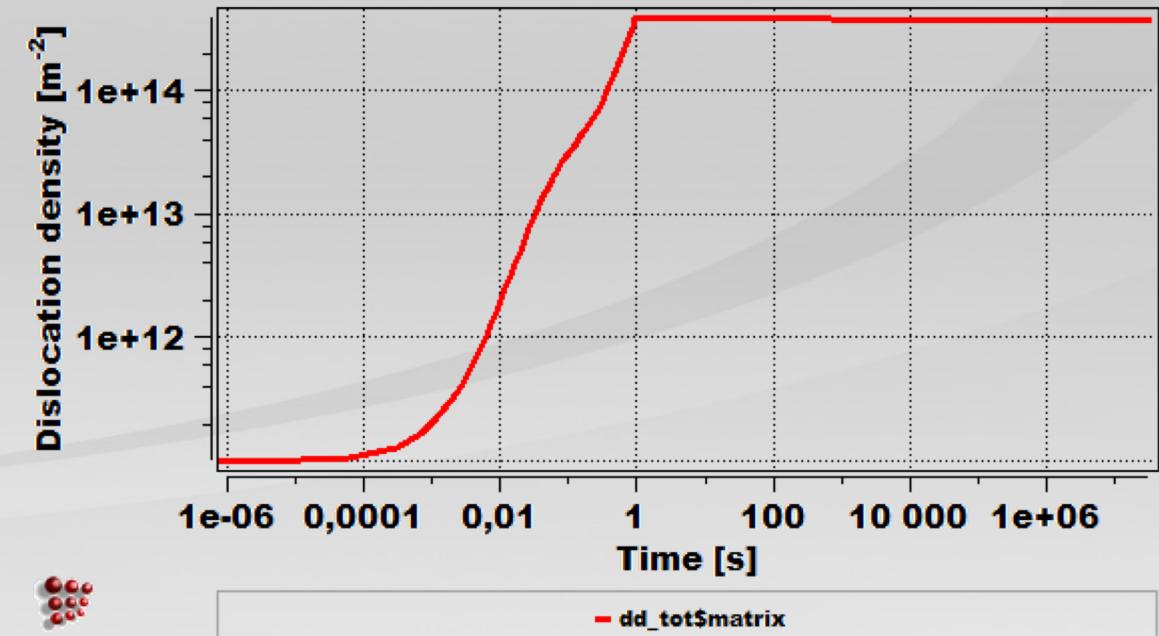
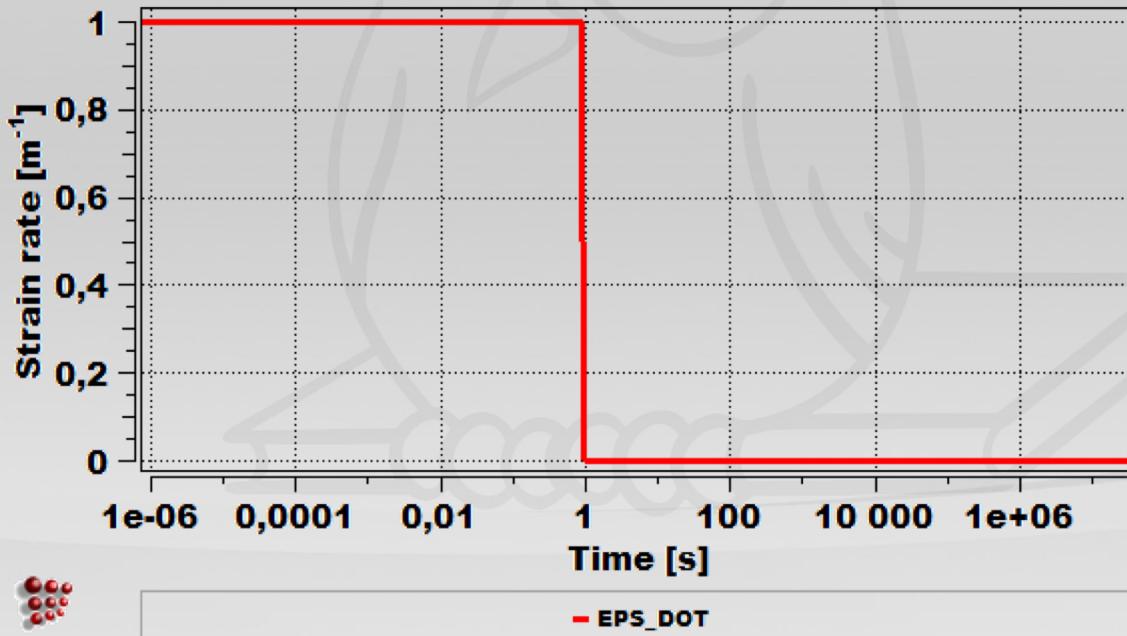
Al, 400°C isothermal, A=50, B=5, C=1e-3 (default values)

1-parameter model



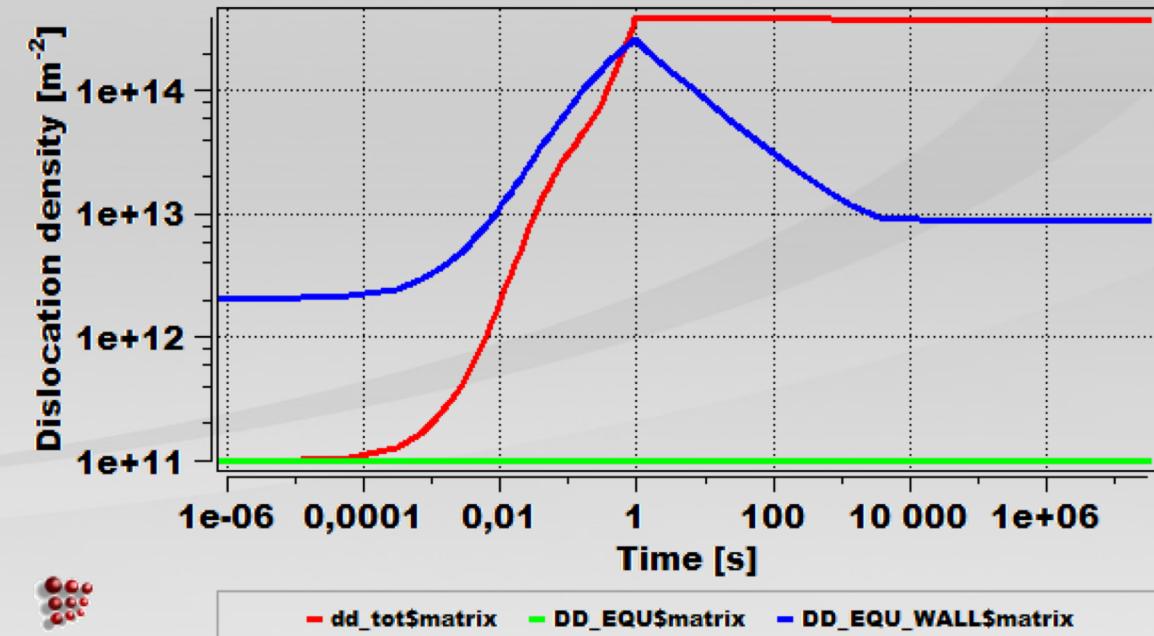
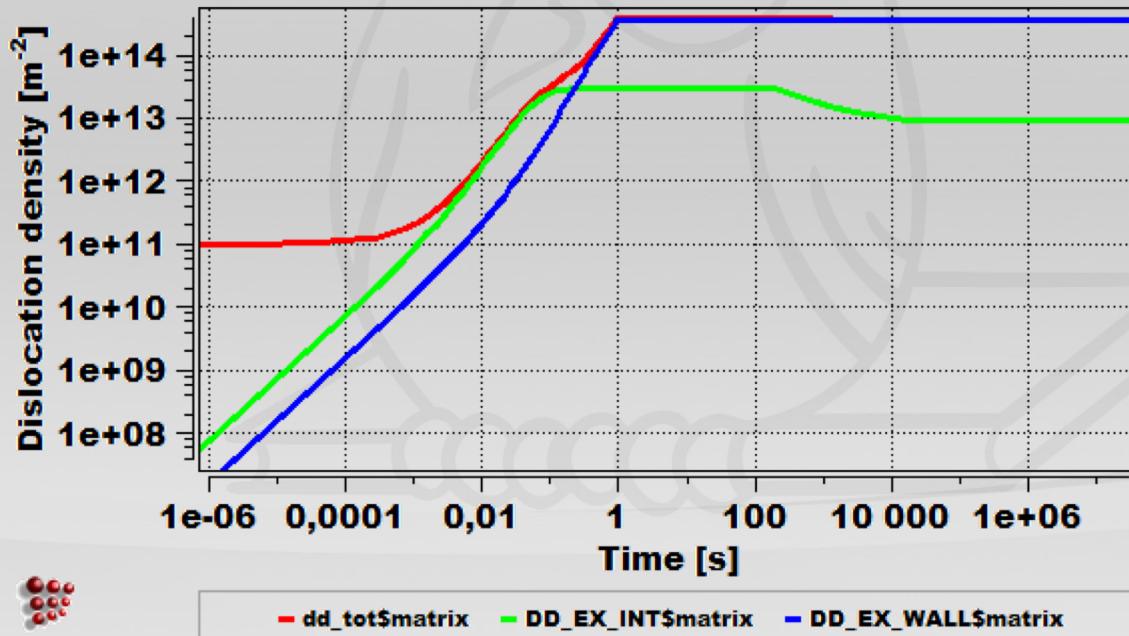
Al, 400°C isothermal, A=50, B=5, C=1e-3 (default values)

2-parameters model



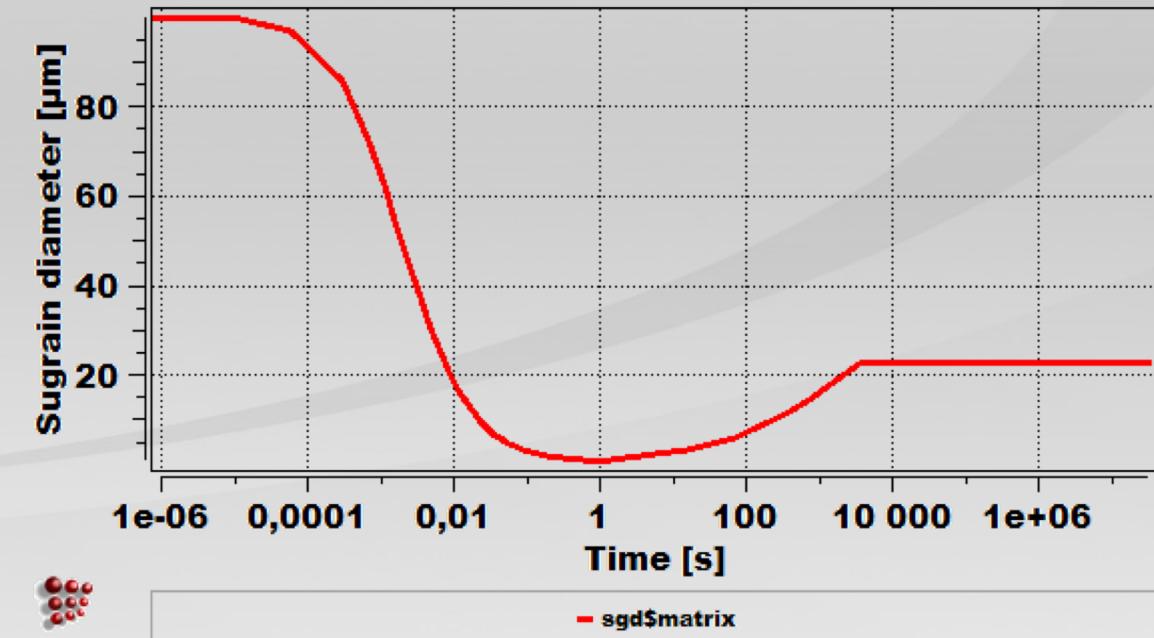
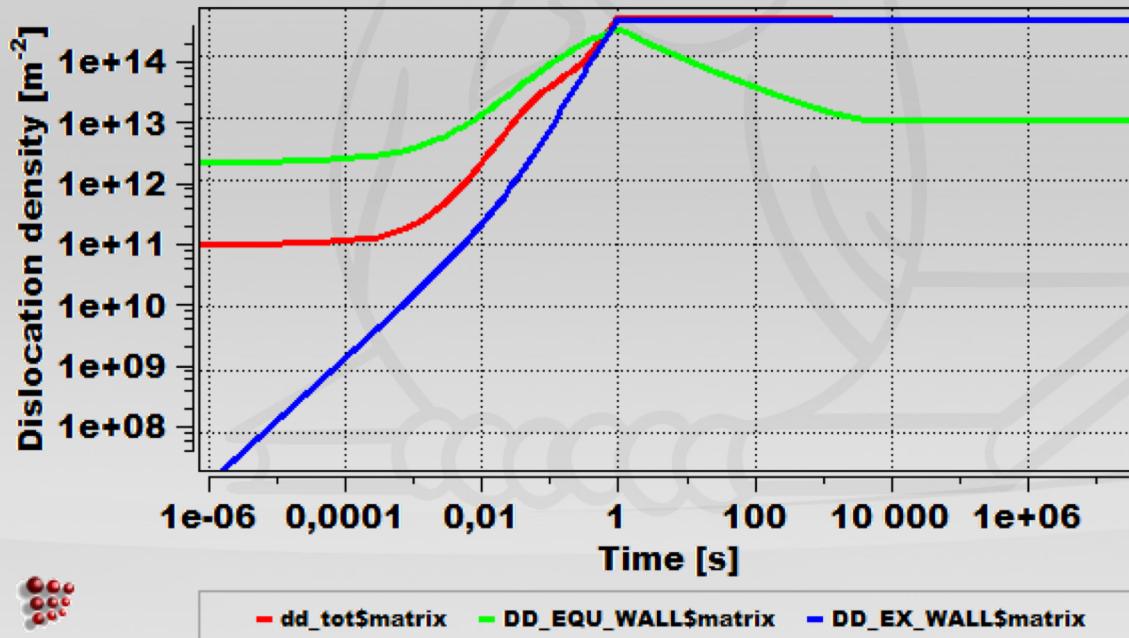
Al, 400°C isothermal, A=50, B=5, C=1e-3, Aw=250, Bw=2e-1, Cw=0 (default values)

2-parameters model



Al, 400°C isothermal, A=50, B=5, C=1e-3, Aw=250, Bw=2e-1, Cw=0 (default values)

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Acknowledgments

- Yao Shan
- Heinrich Buken